Supercooled liquids and the glass transition

Pablo G. Debenedetti* & Frank H. Stillinger†‡

*Department of Chemical Engineering and ‡Princeton Materials Institute, Princeton University, Princeton, New Jersey 08544, USA
(e-mail: pdebene@princeton.edu)
†Bell Laboratories, Lucent Technologies, Murray Hill, New Jersey 07974, USA

Glasses are disordered materials that lack the periodicity of crystals but behave mechanically like solids. The most common way of making a glass is by cooling a viscous liquid fast enough to avoid crystallization. Although this route to the vitreous state — supercooling — has been known for millennia, the molecular processes by which liquids acquire amorphous rigidity upon cooling are not fully understood. Here we discuss current theoretical knowledge of the manner in which intermolecular forces give rise to complex behaviour in supercooled liquids and glasses. An intriguing aspect of this behaviour is the apparent connection between dynamics and thermodynamics. The multidimensional potential energy surface as a function of particle coordinates (the energy landscape) offers a convenient viewpoint for the analysis and interpretation of supercooling and glass-formation phenomena. That much of this analysis is at present largely qualitative reflects the fact that precise computations of how viscous liquids sample their landscape have become possible only recently.

The glassy state is ubiquitous in nature and technology. It is crucial in the processing of foods, the commercial stabilization of labile biochemicals, and the preservation of insect life under extremes of cold or dehydration. Window glass, composed mostly of sand, lime and soda, is the best-known example of an engineered amorphous solid. Optical fibres are made of very pure amorphous silica, occasionally carefully doped. Most engineering plastics are amorphous solids, as are some metallic glasses and alloys of interest because of their soft magnetism and corrosion resistance. The silicon used in many photovoltaic cells is amorphous, and it is possible that most water in the Universe may be glassy. Most of these examples entail supercooling of a liquid to take advantage of viscous retardation of nucleation and crystallization. Understanding quantitatively the extraordinary viscous slow-down that accompanies supercooling and glass formation is a major scientific challenge.

We begin by reviewing the phenomenology of vitrification and supercooling. A useful approach for relating this complex phenomenology to molecular-scale events is to focus attention on the liquid’s energy landscape, that is, the multidimensional surface generated by the system’s potential energy as a function of molecular coordinates. Accordingly, basic landscape concepts and a discussion of the important theoretical and computational progress currently being made in this area are presented next. This is followed by a discussion of alternative viewpoints, in which narrowly avoided singularities are assumed to occur well above the glass-transition temperature. We then close with a summary of the important open questions.

It is impossible to do justice to the entire field of supercooled liquids and amorphous solids in an article of this length. We have therefore limited the scope to the dynamics and thermodynamics of viscous liquids above and close to the glass-transition temperature \( T_g \). — in other words, to the glass transition viewed ‘from the liquid.’ The view ‘from the solid’, including such topics as relaxation both relatively near (for example, during annealing or ageing) and far below \( T_g \) is not discussed. The reader is referred to an excellent recent review for a thorough coverage of these and other topics.

**Phenomenology of supercooling and glass formation**

Figure 1 illustrates the temperature dependence of a liquid’s volume (or enthalpy) at constant pressure. Upon cooling below the freezing point \( T_m \), molecular motion slows down. If the liquid is cooled sufficiently fast, crystallization can be avoided. Eventually molecules will rearrange so slowly that they cannot adequately sample configurations in the available time allowed by the cooling rate. The liquid’s structure therefore appears ‘frozen’ on the laboratory timescale (for example, minutes). This falling out of equilibrium occurs across a narrow transformation range where the characteristic molecular relaxation time becomes of the order of 100 seconds, and the rate of change of volume or enthalpy with respect to temperature decreases abruptly (but continuously) to a value comparable to that of a crystalline solid. The resulting material is a glass. The intersection of the liquid and vitreous portions of the volume versus temperature curve provides one definition of \( T_g \) which usually occurs around \( 2T_m/3 \).

The slower a liquid is cooled, the longer the time available for configurational sampling at each temperature, and hence the colder it can become before falling out of liquid-state equilibrium. Consequently, \( T_g \) increases with cooling rate. The properties of a glass, therefore, depend on the process by which it is formed. In practice, the dependence of \( T_g \) on the cooling rate is weak (\( T_g \) changes by \( 3-5 \text{ °C} \) before falling out of equilibrium). Consequently, \( T_g \) is increases with cooling rate. The properties of a glass, therefore, depend on the process by which it is formed. In practice, the dependence of \( T_g \) on the cooling rate is weak (\( T_g \) changes by \( 3-5 \text{ °C} \) before falling out of equilibrium). Consequently, \( T_g \) is increases with cooling rate. The properties of a glass, therefore, depend on the process by which it is formed. In practice, the dependence of \( T_g \) on the cooling rate is weak (\( T_g \) changes by \( 3-5 \text{ °C} \) before falling out of equilibrium). Consequently, \( T_g \) is increases with cooling rate. The properties of a glass, therefore, depend on the process by which it is formed. In practice, the dependence of \( T_g \) on the cooling rate is weak (\( T_g \) changes by \( 3-5 \text{ °C} \) before falling out of equilibrium).

**Slowing down**

Another definition of \( T_g \) is the temperature at which the shear viscosity reaches \( 10^{13} \text{ poise} \). Close to \( T_g \) the viscosity \( \eta \)
is extraordinarily sensitive to temperature. For silica this dependence is reasonably well described by the Arrhenius functionality,
\[ \eta = A \exp\left[\frac{B}{(T - T_c)}\right] \]
where \( A \) and \( B \) are temperature-independent constants. Understanding the origin of this extraordinary slow-down of relaxation processes is one of the main challenges in the physics of glasses.

Figure 2 shows a \( T_g \)-scaled Arrhenius representation of liquid viscosities. Angell has proposed a useful classification of liquids along a ‘strong’ to ‘fragile’ scale. The viscosity and relaxation times (for example, dielectric relaxation) of the former behave in nearly Arrhenius fashion, whereas fragile liquids show marked deviations from Arrhenius behaviour. Silica (SiO\(_2\)) is the prototypical strong liquid, whereas \( \alpha \)-terphenyl (OTP) is the canonical fragile glass-former. Strong liquids, such as the network oxides SiO\(_2\) and germanium dioxide (GeO\(_2\)), have tetrahedrally coordinated structures, whereas the molecules of fragile liquids exert largely non-directional, dispersive forces on each other. Alternative scaling descriptions that attempt to extract universal aspects of viscous slow-down have been proposed. Their relative merits are still being assessed.

Viscous liquids close to \( T_g \) exhibit non-exponential relaxation. The temporal behaviour of the response function \( F(t) \) (for example, the polarization in response to an applied electric field, the strain (deformation) resulting from an applied stress, or the stress in the polarization in response to an applied electric field, the strain (deformation) resulting from an applied stress, or the stress in the}
response to an imposed deformation) can often be described by the stretched exponential, or Kohlrausch–Williams–Watts (KWW) function\textsuperscript{26,27,38,39}

\[
F(t) = \exp[-((t/\tau)^\beta) \quad (\beta < 1)]
\]

where \(F(t) = [\psi_1(t) - \psi_1(\infty)]/[\psi_1(0) - \psi_1(\infty)]\) and \(\psi_1\) is the measured quantity (for example, the instantaneous stress following a step change in deformation). \(\tau\) in equation (2) is a characteristic relaxation time, whose temperature dependence is often non-Arrhenius (exhibiting fragile behaviour). The slowing down of long-time relaxation embodied in equation (2) contrasts with the behaviour of liquids above the melting point, which is characterized by simple exponential relaxation. Experimental and computational evidence indicates that this slow-down is related to the growth of distinct relaxing domains\textsuperscript{30–39} (spatial heterogeneity). Whether each of these spatially heterogeneous domains relaxes exponentially or not is a matter of considerable current interest\textsuperscript{38,39}.

**Decouplings**

In supercooled liquids below approximately 1.2\(T_g\) there occurs a decoupling between translational diffusion and viscosity, and between rotational and translational diffusion\textsuperscript{30,39,40}. At higher temperatures, both the translational and the rotational diffusion coefficients are inversely proportional to the viscosity, in agreement with the Stokes–Einstein and Debye equations, respectively. Below approximately 1.2\(T_g\), the inverse relationship between rotational motion and viscosity breaks down, whereas that between rotational motion and viscosity does not. Near \(T_g\), it is found that molecules translate faster than expected based on their viscosity, by as much as two orders of magnitude. This therefore means that, as the temperature is lowered, molecules on average translate progressively more for every rotation they execute. Yet another decoupling occurs in the moderately supercooled range. At sufficiently high temperature the liquid shows a single peak relaxation frequency (Fig. 3), indicative of one relaxation mechanism. In the moderately supercooled regime, however, the peak splits into slow (\(\alpha\)) and fast (\(\beta\)) relaxations\textsuperscript{41–43}. The former exhibit non-Arrhenius behaviour and disappear at \(T_g\); the latter continue below \(T_g\) and display Arrhenius behaviour\textsuperscript{44}.

**Thermodynamics**

The entropy of a liquid at its melting temperature is higher than that of the corresponding crystal. Because the heat capacity of a liquid is higher than that of the crystal, this entropy difference increases upon supercooling (Box 1). Figure 4 shows the temperature dependence of the entropy difference between several supercooled liquids and their stable crystals\textsuperscript{35}. For lactic acid this entropic surplus is consumed so fast that a modest extrapolation of experimental data predicts its impending vanishing. In practice, the glass transition intervenes, and \(\Delta S\) does not vanish. If the glass transition did not intervene, the liquid entropy would equal the crystal’s entropy at a nonzero temperature \(T_g\) (the Kauzmann temperature). Because the entropy of the crystal approaches zero as \(T\) tends to zero, the entropy of the liquid would eventually become negative upon cooling if this trend were to continue. Because entropy is an inherently non-negative quantity (Box 1), the state of affairs to which liquids such as lactic acid are tending is an entropy crisis\textsuperscript{46–48}. The extrapolation needed to provoke conflict with the third law is quite modest for many fragile liquids\textsuperscript{34}, and the imminent crisis is thwarted by a kinetic phenomenon, the glass transition. This suggests a connection between the kinetics and the thermodynamics of glasses\textsuperscript{45}. The thermodynamic viewpoint that emerges from this analysis considers the laboratory glass transition as a kinetically controlled manifestation of an underlying thermodynamic transition to an ideal glass with a unique configuration.

A formula of Adam and Gibbs\textsuperscript{51} provides a suggestive connection between kinetics and thermodynamics:

\[
t = A\exp(B/T_s)
\]

In this equation, \(t\) is a relaxation time (or, equivalently, the viscosity) and \(A\) and \(B\) are constants. \(T_s\), the configurational entropy, is related to the number of minima of the system’s multidimensional potential energy surface (Box 2). According to the Adam–Gibbs picture, the entropy is an inherently non-negative quantity (Box 1), and the imminent crisis is thwarted by a kinetic phenomenon, the glass transition. This suggests a connection between the kinetics and the thermodynamics of glasses\textsuperscript{47}. The thermodynamic viewpoint that emerges from this analysis considers the laboratory glass transition as a kinetically controlled manifestation of an underlying thermodynamic transition to an ideal glass with a unique configuration.

![Figure 1](https://example.com/fig1.png)

**Figure 1** Temperature dependence of a liquid’s volume or enthalpy at constant pressure. \(T_g\) is the melting temperature. A slow cooling rate produces a glass transition at \(T_g\); a faster cooling rate leads to a glass transition at \(T_f\). The thermal expansion coefficient \(\alpha_v = (\partial \ln v / \partial T)_p\) and the isobaric heat capacity \(c_p = (\partial h / \partial T)_p\) change abruptly but continuously at \(T_g\).

![Figure 2](https://example.com/fig2.png)

**Figure 2** \(T_g\)-scalet Arrhenius representation of liquid viscosities showing Angel’s strong–fragile pattern. Strong liquids exhibit approximately linear (Arrhenius behaviour), indicative of a temperature-independent activation energy \(E = \ln \eta / \partial \ln (1/T)\) = const. Fragile liquids exhibit super-Arrhenius behaviour, their effective activation energy increasing as temperature decreases. (Adapted from refs 9 and 11.)

© 2001 Macmillan Magazines Ltd
Nevertheless, equation (3) describes the relaxation behaviour of deeply supercooled liquids remarkably well. If the difference in heat capacities between a supercooled liquid and its stable crystalline form is inversely proportional to temperature\(^5\), the Adam–Gibbs relation yields the VTF equation, which is mathematically equivalent to the Williams–Landel–Ferry equation for the temperature dependence of viscosity in polymers\(^5\). This transformation is predicated on the assumption that the vibrational entropies of the

**Box 2 Statistics of landscapes**

The complexity of many-body landscapes makes a statistical description inevitable. The quantity of interest is the number of minima of given depth, which is given by\(^1\)

\[
\frac{d\Omega}{d\phi} = C \exp[N\sigma(\phi)]
\]

Here, \(d\Omega\) denotes the number of potential energy minima with depth per particle \(\phi = \Phi/N\) between \(\phi\) and \(\phi + d\phi/2\). C is an \(N\)-independent factor with units of inverse energy, and \(\sigma(\phi)\), also an \(N\)-independent quantity, is a so-called basin enumeration function. Taking the logarithm of the above expression and comparing with Boltzmann’s entropy formula (Box 1), we see that \(\sigma(\phi)\) is the entropy per particle arising from the existence of multiple minima of depth \(\phi\), or, in other words, the configurational entropy.

At low temperatures, it is possible to separate the configurational contribution to thermophysical properties, which arises from the exploration of different basins, from the vibrational component, which arises from thermal motions confined to a given basin\(^7,8\). The Helmholtz free energy \(A\) is then given by

\[
\frac{A}{NkT} = \frac{\phi}{kT} - \sigma(\phi) + \frac{\phi^2}{2kT}
\]

where \(\phi\) is the depth of the basins preferentially sampled at the given temperature, and \(a\) is the vibrational free energy per particle. Thus, the free energy consists of an energetic component that reflects the depth of landscape basins sampled preferentially at the given temperature, an entropic component that accounts for the number of existing basins of a given depth, and a vibrational component. The statistical description of a landscape consists of the basin enumeration function \(\sigma(\phi)\), from which the excitation profile \(\phi(T)\) is obtained through the free-energy minimization condition

\[
\frac{d\sigma}{d\phi} = \frac{1}{kT}
\]

The above equation assumes that \(a\) depends on \(T\), but not on \(\phi\) — that is, all basins have the same mean curvature at their respective minima.

The shape of a given system’s landscape is determined by the density (number of molecules per unit volume, \(N/V\)). Temperature governs the manner in which the landscape is sampled. A different basin enumeration function and excitation profile corresponds to each density. Temperature dictates the point along the enumeration curve and the excitation profile sampled by the system at fixed density (see figure below).

It is possible to construct the basin enumeration function and excitation profile of a system from experimental heat capacity data for the crystal and the supercooled liquid\(^9,10\), and by computer simulation\(^7,8\). In the latter case, the calculations involve determining the probability distribution of inherent structure energies sampled as a function of temperature. These calculations are at the limit of what is presently feasible with available computational power. The enumeration function is often well represented by a parabola, indicative of a gaussian distribution of basins\(^4,7,7,9\). At present it is not understood how the enumeration function deforms with density for a given system (but see ref. 96 for a recent example of such a calculation), or how it depends on molecular architecture. Understanding such questions would provide a direct link between landscape statistics and physical properties. The success of the Adam–Gibbs equation indicates that this link applies also to transport properties such as diffusion and viscosity.
supercooled liquid and its stable crystal are equal. For many fragile glass-formers the VTF temperature of structural arrest, \( T_g \), is very close to \( T_m \) obtained from calorimetric measurements (typically \( 0.9 < T_g/T_m < 1.1 \)). This again indicates a connection between dynamics and thermodynamics not present at higher temperatures. Equally suggestive is the correspondence between kinetic fragilities based on the temperature dependence of the viscosity (see Fig. 2) and thermodynamic fragilities \(^{41} \), based on the temperature dependence of the entropy surplus of the supercooled liquid with respect to its stable crystal.

The energy landscape

A convenient framework for interpreting the complex phenomenology just described is provided by the energy landscape. This is the name generally given to the potential energy function of an N-body system \( \Phi(r_1, ..., r_N) \), where the vectors \( r_i \) comprise position, orientation and vibration coordinates. In condensed phases, whether liquid or solid, every molecule experiences simultaneous interactions with numerous neighbours. Under these conditions it is convenient to consider the full N-body \( \Phi \)-function. The landscape is a multidimensional surface. For the simplest case of \( N \) structureless particles possessing no internal orientational and vibrational degrees of freedom, the landscape is a \((3N+1)\)-dimensional object. Figure 5 is a schematic illustration of an energy landscape. The quantities of interest are the number of potential energy minima (also called inherent structures) of a given depth (Box 2), and the nature of the saddle points separating neighbouring minima. More than 30 years ago, Goldstein articulated a topographic viewpoint of condensed phases \(^{39-42} \) that has come to be known as the energy landscape paradigm. His seminal ideas have since been applied to protein folding \(^{36-44} \), the mechanical properties of glasses \(^{45-48} \), shear-enhanced diffusion \(^{49} \) and the dynamics of supercooled liquids \(^{50-51} \).

Landscape sampling

For an \( N \)-body material system in a volume \( V \), the landscape is fixed. The manner in which a material system samples its landscape as a function of temperature provides information on its dynamic behaviour.\(^{39} \) The way that a landscape deforms as a result of changes in density provides information on the mechanical properties of a material system. \(^{72} \) Figure 6 shows the average inherent structure energy for a mixture of unequal-sized atoms, as a function of the temperature of the equilibrated liquid. \(^{40,73} \) In these calculations, molecular dynamics simulations of the binary mixture were performed to generate configurations. Periodically, the system’s energy was minimized, yielding mechanically stable inherent structures, the average energy of which is reported in the figure. At high temperatures the inherent structure energy is virtually temperature-independent, and appears to have reached a plateau. When the system has sufficient kinetic energy to sample its entire energy landscape, the overwhelming number of minima that it samples are shallow, reflecting the fact that deep minima are very rare (Box 2). But as the reduced temperature decreases below about \( T = 1 \), the system is unable to surmount the highest energy barriers, and is therefore forced to sample the much rarer deeper minima (Box 2). When this happens, the kinetics of structural relaxation changes from exponential to stretched exponential, and the activation energy (and entropy) associated with structural relaxation become super-Arrhenius, that is to say they increase with decreasing temperature.\(^{50} \)

These calculations established a connection between changes in dynamics and the manner in which the static and thermodynamic energy landscape is sampled as a function of temperature. Figure 6 also shows that at a low enough temperature the system becomes stuck in a single minimum, the depth of which increases as the cooling rate decreases. This corresponds to the glass transition. Another important observation of this study was the existence of a temperature \( T = 0.45 \), below which the height of the barriers separating sampled inherent structures increases abruptly. This temperature was found to correspond closely to the crossover temperature predicted by mode-coupling theory (MCT; see below) for this system. Here again, it is the manner in which the system samples its landscape, not the landscape itself, that changes with temperature. (See ref. 74 for a recent, different interpretation of landscape sampling at this temperature.)

The landscape picture provides a natural separation of low-temperature molecular motion into sampling distinct potential
energy minima, and vibration within a minimum. It is possible to separate formally the corresponding configurational and vibrational contributions to a liquid’s properties\(^{58,59}\). In two important computational studies, the configurational entropy was calculated by probing systematically the statistics governing the sampling of potential energy minima\(^{77,78}\) (Box 2). Using this technique, a remarkable connection between configurational entropy and diffusion was identified in liquid water\(^{80}\). One of water’s distinguishing anomalies is the fact that, at sufficiently low temperature, its diffusivity increases upon compression\(^{80}\). As shown in Fig. 7, diffusivity maxima are correlated strongly with configurational entropy maxima, the respective loci coinciding within numerical error.

The results shown in Fig. 7 and the success of the Adam–Gibbs equation in describing experimental data on relaxation in a wide variety of systems\(^{52}\) indicate that there exists a scaling relationship between the depth distribution of basins and the height of the saddle points along paths connecting neighbouring basins. Such scaling is not a mathematical necessity, but arises from the nature of real molecular interactions. The topographic nature of this statistical scaling relationship between minima and saddle points is poorly understood (but see the recent computational investigation of saddle points\(^{42}\)). Its elucidation will explain the origin of the connection between the dynamics and thermodynamics of glass-forming liquids, and constitutes the principal theoretical challenge in this field.

**Strong versus fragile behaviour**

The extent to which the shear viscosity \(\eta\) deviates from Arrhenius behaviour, \(\eta = \eta_0 \exp(E/RT)\), constitutes the basis of the classification of liquids as either strong or fragile (Fig. 2). Molten SiO\(_2\), often considered as the prototypical strong glass-former, displays an almost constant activation energy of 180 kcal mol\(^{-1}\) (ref. 81). This constancy indicates that the underlying mechanism, presumably breaking and reformation of Si–O bonds, applies throughout the entire landscape\(^4\). In contrast, the viscosity of OTP — the canonical fragile glass-former — deviates markedly from Arrhenius behaviour\(^{42}\), showing an effective activation energy \((\text{dln}\eta/\text{dT})/T\) that increases 20-fold, from one-quarter of the heat of vaporization for the liquid above its melting point to roughly five times the heat of vaporization near \(T_g\). This means that OTP’s landscape is very heterogeneous. The basins sampled at high temperature allow relaxation by surmounting low barriers involving the rearrangement of a small number of molecules. The very large activation energy at \(T = T_g\), on the other hand, corresponds to the cooperative rearrangement of many molecules. These differences between strong and fragile behaviour imply a corresponding topographic distinction between the two archetypal landscapes. Aside from multiplicity due to permutational symmetry, strong landscapes may consist of a single ‘megabasin’, whereas fragile ones display a proliferation of well-separated ‘megabasins’ (Fig. 8).

Cooperative rearrangements such as those that must occur in OTP are unlikely to consist of elementary transitions between adjacent basins. Rather, the likely scenario involves a complicated sequence of elementary transitions. At low temperatures, these rearrangements should be rare and long-lived on the molecular timescale. Furthermore, the diversity of deep landscape traps and of the pathways of configuration space that connect them should result in a broad spectrum of relaxation times, as required for the stretched exponential function in equation (2). This in turn suggests that supercooled fragile liquids are dynamically heterogeneous, probably consisting at any instant of mostly non-diffusing molecules with a few ‘hot spots’ of mobile molecules. This dynamic heterogeneity\(^9\) has both experimental\(^{30,35,37}\) and computational\(^{39,43,45}\) support.

The inverse relation between the self-diffusion coefficient and viscosity embodied in the Stokes–Einstein equation is based on macroscopic hydrodynamics that treats the liquid as a continuum.

**Figure 5** Schematic illustration of an energy landscape. The x-axis represents all configurational coordinates. (Adapted from ref. 44.)

**Figure 6** Mean inherent structure energy per particle of a binary mixture of unequal-sized Lennard–Jones atoms, as a function of the temperature of the equilibrated liquid from which the inherent structures were generated by energy minimization. Molecular dynamics simulations at constant energy and density were performed over a range of temperatures for 256 Lennard–Jones atoms, of which 20% are of type A and 80% are of type B. The Lennard–Jones size and energy parameters are \(\sigma_{AA} = 1\), \(\sigma_{AB} = 0.88\), \(\sigma_{BB} = 0.8\), and \(\epsilon_{AA} = 1\), \(\epsilon_{AB} = 0.5\), \(\epsilon_{BB} = 1.5\), respectively. Length, temperature, energy and time are expressed in units of \(\sigma_{AA}E_{AA}/\epsilon_{AA}\), \(\sigma_{BB}E_{BB}/\epsilon_{BB}\) and \(\sigma_{AA}E_{AA}/\epsilon_{AA}\), respectively, with \(m\) representing the mass of the particles. Simulations were performed at a density of 1.2. The fast and slow cooling rates are \(1.08 \times 10^{-3}\) and \(3.33 \times 10^{-4}\) when \(T > 1\). When \(T > 1\), the system has sufficient kinetic energy to sample the entire energy landscape, and the overwhelming number of sampled energy minima are shallow. Under these conditions, the system exhibits a temperature-independent activation energy for structural relaxation (calculations not shown). Between \(T = 1\) and \(T = 0.45\), the activation energy increases upon cooling, the dynamics become ‘landscape-influenced’, and the mechanically stable configurations sampled are strongly temperature-dependent. Below \(T = 0.45\), the height of the barriers separating sampled adjacent energy minima seems to increase abruptly (calculations not shown). This is the ‘landscape-dominated’ regime. In it, particles execute rare jumps over distances roughly equal to interparticle separations. The crossover between landscape-influenced and landscape-dominated behaviour corresponds closely with the mode-coupling transition temperature\(^{23,25}\). (Adapted from refs 70 and 72.)
This picture must clearly break down in supercooled fragile liquids, which are dynamically heterogeneous. The failure of the Stokes–Einstein equation, referred to above as one of the distinguishing characteristics of fragile supercooled liquids, is therefore qualitatively understandable. Plausible models for the low-temperature enhancement of diffusive motion relative to hydrodynamic expectations based on the viscosity have been proposed, but an accurate predictive theory is missing. The landscape viewpoint also provides a plausible interpretation for the $\alpha/\beta$-relaxation decoupling shown in Fig. 3 — $\alpha$-relaxations correspond to configurational sampling of neighboring megabasins (Fig. 8), whereas $\beta$-processes are thought to correspond to elementary relaxations between contiguous basins. Direct computational evidence of this interpretation is not available.

**Avoided singularities**

Alternative viewpoints to the landscape perspective have also contributed to current understanding of some aspects of supercooling and the glass transition. Two such interpretations invoke a narrowly avoided singularity above $T_x$.

According to MCT, structural arrest occurs as a result of the following feedback mechanism: (i) shear-stress relaxation occurs primarily through diffusive motion; (ii) diffusion and viscosity are inversely related; and (iii) viscosity is proportional to shear-stress relaxation time. These facts lead to a viscosity feedback whereby structural arrest occurs as a purely dynamic singularity, that is to say it is not accompanied by thermodynamic signatures such as a diverging correlation length. What is now known as the idealized MCT predicts structural arrest to occur at a temperature $T_x$. Initially, therefore, it was thought that MCT was a useful theory for the laboratory-generated glass transition. It is now widely understood that this is not the case, as one finds that $T_x > T_\alpha$, and the MCT-predicted singularity does not occur. In subsequent modifications of the theory, additional relaxation mechanisms occur, often referred to as ‘hopping’ or activated motions, which restore ergodicity (the system’s ability to sample all configurations) below $T_x$, thereby avoiding a kinetic singularity. These additional relaxation modes arise as a result of a coupling between fluctuations in density and momentum.

Although not a theory of the glass transition, MCT accurately describes many important aspects of relaxation dynamics in liquids above or moderately below their melting temperatures. In particular, the theory makes detailed predictions about the behaviour of the intermediate scattering function $F$, an experimentally observable quantity that measures the decay of density fluctuations. After a fast initial decay due to microscopic intermolecular collisions, MCT predicts that the decay of $F$ obeys the following sequence (Fig. 9): (i) power-law decay towards a plateau, according to $F \propto f + A r^{-t}$; (ii) a second power-law decay away from the plateau value $F \propto f - B r^{t}$; and (iii) slow relaxation at longer times, which can be fitted by the KWW function $F \propto \exp\left[-(t/\tau)^b\right]$. Here, $f$ is the plateau value of the scattering function, which only appears at sufficiently low temperature; $t$ is time; $A$, $B$, $a$ and $b$ are constants; $\tau$ is the characteristic, temperature-dependent relaxation time; and $B < 1$ is the KWW stretch exponent. The basic accuracy of these detailed predictions has been verified experimentally and in computer simulations.

Kivelson and co-workers have proposed a theory of supercooled liquids that is based also on an avoided singularity. According to this viewpoint, the liquid has an energetically preferred local structure that differs from the structure in the actual crystalline phase. The system is prevented from crystallizing into a reference crystal with the preferred local structure because of geometric frustration owing to the fact that the latter does not tile space. An example of such energetically favoured but non-space-tiling local structure is the icosahedral packing seen in computer simulations of the supercooled Lennard–Jones liquid. At a temperature $T^*$ the system would, but for frustration, crystallize into the reference crystal. Instead, strain build-up causes the system to break up into frustration-limited domains, thereby avoiding a phase transition (singularity) at $T^*$. The avoided transition temperature $T^*$ acts as a critical point, below which two length scales emerge, both of which are large compared to
molecular dimensions. One is the critical correlation length, which governs density fluctuations in the absence of frustration. The second is the frustration-limited domain size. From these considerations there emerge predictions on the temperature dependence of the viscosity. Experimental data analysed according to the theory display universality\(^9\), but at the expense of introducing a number of fitting parameters. The improvement with respect to theoretical models\(^9,7\) is therefore of paramount importance.

**Challenges and open questions**

Important aspects of the complex behaviour of viscous liquids close to the glass transition can be explained qualitatively from the energy landscape perspective. Making this descriptive picture quantitative and predictive is a major challenge. This will require investigating how basic landscape features such as the basin enumeration function depend on molecular architecture and, for a given substance or mixture, on density (see ref. 96 for an example of such a calculation). Equally important is the translation of qualitative pictures such as Fig. 8 into precise measures of strength and fragility based on the basin enumeration function. Uncovering the topographic nature of the scaling relationship between basin minima and saddle points holds the key to understanding the relationship between kinetics and thermodynamics in deeply supercooled liquids. All of these calculations are in principle straightforward, but computational at the very limit of what is currently feasible. The development of theoretical models\(^9\) is therefore of paramount importance.

MCT and the landscape perspective offer complementary view-points of the same phenomena. So far, however, not enough effort has been devoted to bridging the gap that separates these two approaches. Recent calculations\(^8,9\) offer the promise of establishing a clearer connection between the static landscape viewpoint and the dynamic perspective of MCT. At the least, what is required is a precise landscape-based explanation of what ‘hopping’ and ‘activated processes’ really mean. Additional theoretical viewpoints of supercooling and the glass transition include the instantaneous normal-mode perspective on liquid dynamics\(^9\) and thermodynamic treatments of the vitreous state based on invoking analogies to spin glasses\(^{10}\). Establishing a coherent theoretical perspective on supercooled liquids and glasses is important. We believe that the landscape formalism offers the natural technical tools for accomplishing this task.

---
